

SIMS Investigation of Oxygen in 3C-SiC on Si

Author

Han, Jisheng, Dimitrijević, Sima, Kong, Frederick, Atanacio, Amanda

Published

2009

Journal Title

Journal of Materials Science and Engineering

Rights statement

© 2009 David Publishing. The attached file is reproduced here in accordance with the copyright policy of the publisher. Please refer to the journal's website for access to the definitive, published version.

Downloaded from

<http://hdl.handle.net/10072/34741>

Link to published version

http://davidpublishing.org/journals_info.asp?jId=873

Griffith Research Online

<https://research-repository.griffith.edu.au>

SIMS investigation on the 3C-SiC on Si*

HAN Ji-sheng¹, Sima Dimitrijević¹, Fred Kong¹, Armand Atanacio²

(1. Queensland Microtechnology Facility and Griffith School of Engineering, Griffith University, Nathan, Qld. 4111, Australia;

2. Australian Nuclear Science and Technology Organization, Lucas Heights NSW 2234, Australia)

Abstract: In this paper, the spectrometry (SIMS) measurements of oxygen concentration in 3C SiC epitaxial layers on Si were presented and analysed. The concentration of oxygen determined by SIMS was as high as 10^{19} to 10^{20} atom/cm³. Unlike silicon, oxygen can act as donor atoms in SiC with calculated ionization levels of 200 meV^[1-2]. It is generally believed that the main contribution of dopant concentration in the unintentionally doped SiC film is related to background nitrogen. Because of the high ionisation level, oxygen is not electrically active at room temperature. By measuring the conductivity of the films at higher temperatures, we extracted three donor energy levels: $E_{A1} = 79$ meV, $E_{A2} = 180$ meV, and $E_{A3} = 350$ meV. The activation energy of 180 meV could be associated with the calculated ionization level for oxygen. Further analysis of the conductivity measurements at elevated temperatures will be performed to determine the electrically active donor concentration that is associated with the activation energy of 180 meV.

Key words: 3C-SiC; SIMS; oxygen concentration; nitrogen concentration

1. Introduction

Cubic silicon carbide (3C-SiC) is considered to be excellent material for electronic devices operating at harsh environment because of its high thermal conductivity, high critical electric field, high saturation velocity and wide band gap (2.2 eV). Using the chemical vapor deposition techniques, 3C-SiC on silicon (3C-SiC/Si) wafers with large diameters can be achieved and at low cost^[3-4]. It is generally believed that the main contribution of dopant concentration in the unintentionally doped SiC film is related to

background nitrogen. Recently, Eickoff, et al^[5] observed a deep donor level with ionization energy level of 195 meV in undoped 3C-SiC layers deposited on the SOI substrate from the temperature dependence of the carrier concentration. They attributed the origin of deep donor to the oxygen. The question of how the oxygen affect the electrical properties in 3C-SiC films has not been fully addressed. In this paper, the Secondary Ion Mass Spectrometry (SIMS) measurements of oxygen concentration in 3C-SiC epitaxial layers on silicon were presented and analysed. And the measurements results and analyses of conductivity at different temperatures in the corresponding 3C-SiC films were presented.

2. Experiment

The investigated 3C-SiC films were prepared on silicon substrate at 1000°C deposition temperature using ALE (Atomic Layer Epitaxy) technique^[6-7]. Table 1 shows different growth conditions of the investigated 3C-SiC films. The growth rate of the investigated 3C-SiC films are around 0.5-0.8 nm/cycle. SIMS measurements were conducted at two different laboratories: Australian Nuclear Science and Technology Organisation (ANSTO) Sydney, and Evans Analytical Group (EAG), USA. Some issues remain with SIMS analysis such as nitrogen has very high ionisation potential (14.5 eV), therefore ion yield is very low during SIMS analysis. Better ion yield is obtained from the cluster ($C^{12}N^{14}$). EAG labs used a very high ion beam current and claim to be able to determine accurate volume concentrations of N in SiC

* **Acknowledgment:** This work was supported by an Australian Institute of Nuclear Science and Engineering (AINSE) and by Qs Semiconductors.

Corresponding author: HAN Ji-sheng (1962-), male, Ph.D.; research fields: semiconductor and microelectronics. E-mail: j.han@griffith.edu.au.

with lower limits of 2×10^{15} atoms/cm³[8]. The SIMS experiments in EAG were performed using a CAMECA IMS-4f double focusing magnetic sector instruments equipped with oxygen and Cs primary ion beam sources, which is rastered over a square area. Beam current used in the analysis are 50 to 280 nA. The experiments in ANSTO were performed using a CAMECA IMS-5f double focusing magnetic sector instruments equipped with oxygen and Cs primary ion

beam sources. Beam current used in the analysis were 50 to 230 nA and rastered over a square area 250×250 μm . The electrical properties of the 3C-SiC films were also investigated for the temperature range 300 K to 550 K using standard four-point probe technique without removal substrates. Ohmic contacts for four-point probe test structures were prepared by evaporation of Ni.

Table 1 Different growth conditions of 3C-SiC films

Sample No.	ALE films (1000°C)	SiC film thickness
F1	SiH ₄ flow, pump out then C ₃ H ₆ flow	500 nm
F2	SiH ₄ and C ₂ H ₂ flow together	525 nm
F3	SiH ₄ flow, pump out then C ₂ H ₂ flow	352 nm

3. Results and discussion

Fig. 1 shows the depth profiles of C, N, and O for different 3C-SiC samples using the C¹²N¹⁴ cluster ion obtained from SIMS measurement at ANSTO. The result of sample F1 and F3 which were prepared with different epitaxy conditions of SiC on Si shows that high O counts were both detected which indicates that a high chemical concentration of oxygen exists in the samples. To confirm the high oxygen counts of the samples, an individual SIMS measurement was performed on a commercial Si substrate and included in Fig. 1. The result shows that high oxygen counts were also detected which indicates that a high chemical concentration of oxygen exists in both the Si substrate and SiC. High level of oxygen (up to $10^{20}/\text{cm}^3$) in the sample F2 was detected by EAG SIMS measurement as shown in Fig. 2. We have also extracted donor ionisation energies of the different samples from the conductivity measurement. The results are listed in Table 2.

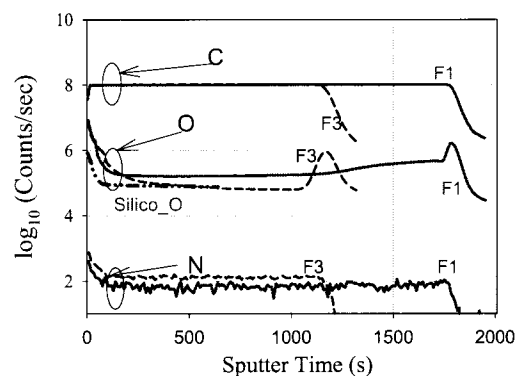


Fig. 1 Negative ions concentration profile in 3C-SiC samples measured by SIMS at ANSTO

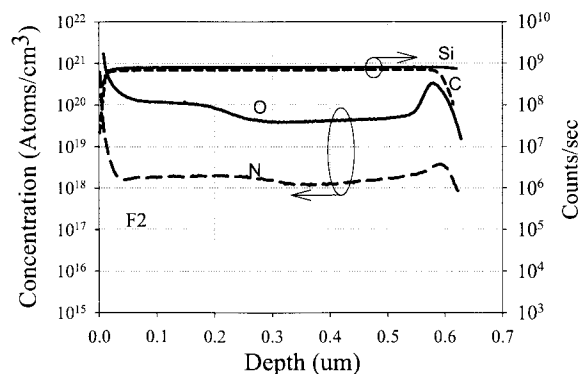


Fig. 2 Negative ions concentration profile in 3C-SiC samples measured by SIMS at EAG

Table 2 Extracted activation energies for different 3C-SiC films

	E_{A1} (eV)	E_{A2} (eV)	E_{A3} (eV)
F1	0.057	0.13	-
F2	0.060	0.18	-
F3	0.079	0.18	0.35

Generally, all the films investigated exhibit shallow donor ionization energy, E_{A1} , of less than 80meV from room temperature to 120°C. The energy levels extracted are similar to nitrogen levels as determined by others^[9-10] as Fig. 3 shows the temperature dependence of the electrical conductivity of the sample.

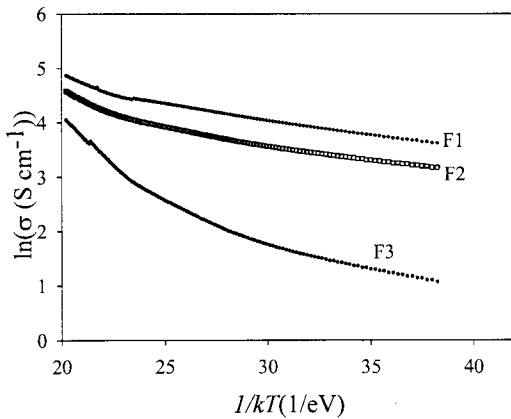


Fig. 3 Measured conductivity versus temperature for different samples

In fact, O and N are two potential elements as donors to contribute to our measured conductivity. Both conductivity measurement and extracted activation energy indicate that O and N are two elements as donors. Although SIMS measurements show a higher count of O in sample F1 compared to F3, the extracted activation energy of F1 indicates that nitrogen is still the main contribution in carrier concentration. However, the extracted activation energy of F1 starts to show as 0.180 eV at higher (> 120°C) temperature which is close to the calculated ionization energy. This result indicates that O is partly involved in the measured conductivity also.

Therefore, high chemical concentration oxygen does not necessarily mean that all the oxygen in the deposited film is electrically active. Electrical conductivity measurement shows that sample F3 exhibits another deeper donor level, $E_{A3} = 350$ meV. The origin of this donor is not clear at the moment and it is likely due to structural defects at the 3C-SiC/Si interface. We also cannot rule out that there is no parallel conduction due to the carrier from the silicon substrate at this high temperature. As for Si, generally there is no significant change on temperature difference on the conductivity. Therefore, unlike SiC, O does not play a major role in Si doping. Instead, N is a major donor contributor for the measured conductivity. The interstitial location of oxygen in the Si crystal might account for this observation-high chemical concentration oxygen and low electrical activity of oxygen.

4. Conclusion

Oxygen was observed in all samples analysed at ANSTO and EAG. High levels of oxygen (up to $10^{20}/\text{cm}^3$) in the sample F2 was detected by EAG SIMS measurement. High oxygen counts were observed also in sample F1 and F3 in both the SiC and Si substrate by ANSTO SIMS measurement. The level in the Si was confirmed by measuring an unprocessed commercial Si wafer. High oxygen levels appeared to contribute to our measured electrical conductivity at elevated temperature and does not degrade crystal quality.

References:

- [1] Gali D., Heringer P., Deak Z., et al. Isolated oxygen defects in 3C- and 4H-SiC: A theoretical study. *Phys. Rev. B*, 2002, 66: 125208.
- [2] Thomas D., Hubert T. G. P., Tsunenobu K., et al. Oxygen in silicon carbide: Shallow donors and deep acceptors. *Mater. Sci. and Eng. B*, 1999, 61-62: 454-459.
- [3] Nishino S., Powell J. A. and Will H. A.. Production of large-area single-crystal wafers of cubic SiC for semiconductor devices. *Appl. Phys. Lett.*, 1983, 42(5): 460-462.

- [4] Nagasawa H. and Yagi K.. 3C-SiC single-crystal films grown on 6-inch Si substrates. *Phys. Stat. Sol. (b)*, 1997, 202(1): 335-358.
- [5] Eickhoff M., Möller H., Stoemenos J., et al. Influence of crystal quality on the electronic properties of n-type 3C-SiC grown by low temperature low pressure chemical vapor deposition. *J. Appl. Phys.*, 2004, 12(12): 7908-7917.
- [6] Matsunami H., Hatayama T. and Fuyuki T.. Hetero-interface control and atomic layer epitaxy of SiC. *Appl. Surf. Sci.*, 1997, 112: 171-175.
- [7] Hatayama T., Tarui Y., Fuyuki, T., et al. Low-temperature heteroepitaxial growth of cubic SiC on Si using hydrocarbon radicals by gas source molecular beam epitaxy. *J. Cryst. Growth.*, 1995, 150: 934-938.
- [8] SIMS detection limits of selected elements in SiC under normal depth profiling conditions. Available at: <http://www.eaglabs.com/publications/appnotes/AN337.pdf>.
- [9] Harris G. L.. *Properties of Silicon Carbide*. INSPEC, London, 1995.
- [10] Ikeda M., Matsunami H. and Tanaka K.. Site effect on the impurity levels in 4H, 6H, and 15R SiC. *Phys. Rev. B.*, 1980, 22: 2842-2854.

(Edited by Malik L. and Donna T.)